magnetization process can be considered as being reversible in temperature. From the magnetization curve at 5 K, the saturation magnetization was determined as 3.9 μB per formula unit. This high magnetization originates from the parallel alignment of the Mn and Fe moments, though the moments of Mn are much larger than those of Fe (ref. 12). Variation of the Mn/Fe ratio may also be used to further improve the MCE.

The magnetic entropy changes, ΔS_m, are calculated from magnetization data by means of the equation

$$\Delta S_m(T, B) = S_m(T, B) - S_m(T, 0) = \int \left( \frac{dM}{dT} \right) dB$$

which is based on a Maxwell relation. The results are shown in Fig. 3. The calculated maximum values of the magnetic entropy changes are 14.5 J kg^{-1} K^{-1} and 18 J kg^{-1} K^{-1} for field changes from 0 to 2 T and 0 to 5 T, respectively. The maximum magnetic entropy in 3d materials depends on the spin moment S. Because there are two magnetic ions per formula unit, we have S_m = 2 ln(2S + 1), where R is the universal gas constant. From the saturation magnetic moment, we estimate the average S value of the magnetic ions to be 1, thus S_m = 18.3 J mol^{-1} K^{-1} = 117 J kg^{-1} K^{-1}, which is about 6 times larger than the value obtained from the magnetization measurements. For comparison, the magnetic entropy change of the compound Gd₅Ge₂Si₂ and Gd is also shown in Fig. 3. It is evident that the MCE in MnFePₓAs₁₋₅ compounds is comparable with that of Gd₅Ge₂Si₂, though the Gd compound has a larger magnetic moment at 5 K. The origin of the large magnetic entropy change is in the comparatively high 3d moments and in the rapid change of magnetization in the field-induced magnetic phase transition. In rare-earth materials, the magnetic moment fully develops only at low temperatures, and therefore the entropy change near room temperature is only a fraction of the potential value. In 3d compounds, the strong magneto-crystalline coupling results in competing intra- and inter-atomic interactions, and leads to a modification of metal–metal distances which may change the iron and manganese magnetic moments and favor spin ordering.

The excellent magnetocaloric features of the compound MnFePₓ₀.₄₅As₀.₅₅, in addition to the very low material costs, make it an attractive candidate material for a commercial magnetic refrigerator.

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**Remote electronic control of DNA hybridization through inductive coupling to an attached metal nanocrystal antenna**

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Increasingly detailed structural1 and dynamic2–5 studies are highlighting the precision with which biomolecules execute often complex tasks at the molecular scale. The efficiency and versatility of these processes have inspired many attempts to mimic or harness them. To date, biomolecules have been used to perform computational operations4 and actuation5, to construct artificial transcriptional loops that behave like simple circuit elements6 and to direct the assembly of nanocrystals8. Further development of these approaches requires new tools for the physical and chemical manipulation of biological systems. Biomolecular activity has been triggered optically through the use of chromophores9–11, but direct electronic control over biomolecular ‘machinery’ in a specific and fully reversible manner has not yet been achieved. Here we demonstrate remote electronic control over the hybridization behaviour of DNA molecules, by inductive coupling of a radio-frequency magnetic field to a metal nanocrystal covalently linked to DNA12. Inductive coupling to the nanocrystal increases the local temperature of the bound DNA, thereby inducing denaturation while leaving surrounding molecules relatively unaffected. Moreover, because dissolved biomolecules disspate heat in less than 50 picoseconds (ref. 16), the switching is fully reversible. Inductive heating of macroscopic samples is widely used17–19, but the present approach should allow extension of this concept to the control of hybridization and thus of a broad range of biological functions on the molecular scale.

Inductive coupling is the transfer of energy between circuits. If the secondary circuit has finite impedance, eddy currents are produced which are converted to heat by the Joule effect17,18. Heating a conductor by placing it in an alternating magnetic field is generally used to heat macroscopic samples. Here we apply it to metallic nanocrystals (diameter 1.4 nm) in solution. Induction heating is accompanied by a skin-depth effect, resulting from partial cancellation of the magnetic fields. Consequently, most of the power absorbed by the conductor is concentrated in a depth d₀ given by:

$$d_0 = \frac{1}{2\pi} \sqrt{\frac{\rho \mu_0 \mu_r f}{\pi}}$$

where $\mu_r$ is magnetic permeability, $\rho$ is the permeability of free space, $\mu_0$ is the material resistivity, and $f$ is the frequency of the alternating magnetic field. The power density $P$ is given by:

$$P = 4\pi H^2_0 \mu_0 \mu_r f d_0$$

where $d$ is the sample diameter, $H_0$ is the magnetic field strength, and $F$ is a transmission factor that has a sigmoidal dependence on d/d₀. Optimal power absorption and heating occur when d/d₀ = 3.5. To optimally heat by induction a gold nanocrystal with d = 1.4 nm, f = 49 GHz (radio frequency range) is required17. Here we use
\[ f = 1 \text{ GHz}, \] which results in a skin depth larger than the particle so that the entire particle is heated. We note that additional mechanisms, including induced dipole torque on the nanoparticle which increases the kinetic energy of the nanoparticles, may be important at these length scales. The physics of heating nanometre particles has not been explored previously, and will be the subject of future investigations. There are numerous examples of the heating of magnetic particles by a magnetic field for the treatment of tumours through macroscopic tissue heating\(^{19}\). However, we emphasize that the (magnetite) particles in such cases are several orders of magnitude larger in volume than the particles used in our investigation, and are not interfaced to individual molecules.

In order to demonstrate reversible electronic control, we have constructed a DNA hairpin-loop oligonucleotide covalently linked to a nanometre scale antenna. 38-nucleotide hairpin-loop DNA which is self-complementary at each end for 7 bases\(^{20}\) (Fig. 1a) was covalently linked to a 1.4-nm gold nanocrystal by a primary amine appended to a single base\(^{21–24}\). Because of the loop constraint, the DNA rehybridizes on a timescale comparable to dehybridization\(^{25}\). The dehybridization was monitored by the hyperchromicity of DNA, measured by optical absorbance at 260 nm ($A_{260}$)\(^{26}\). A solution of the nanocrystal-linked oligonucleotides (referred to as $M$) was put into a radio-frequency magnetic field (RFMF) with $f = 1 \text{ GHz}$. $A_{260}$ was monitored in an ultraviolet–visible spectroscopy.

In Figure 2, we determine the effective temperature from inductive coupling to a gold nanocrystal linked to DNA. (a) The two-phase system. X is a 12-nucleotide DNA molecule covalently labelled with a gold nanocrystal on its 3' end and the fluorophore FAM on its 5' end. Its complementary strand Y has biotin (B) on the 5' end, which was captured onto agarose beads with streptavidin (SA) on the surface. Once X dehybridizes from Y it can diffuse into the supernatant. (b) Supernatant fluorescence spectra. Each sample is subject to a fixed temperature (22°C, 30°C, 50°C, 70°C) or RFMF (solid line). (c) Integrated peak intensity of the supernatant fluorescence spectra shown in (b) (circles) and a sigmoidal fit (solid line), and the intensity of the sample exposed to RFMF (dotted line). Extrapolation of RFMF sample intensity results in an effective temperature of 35°C for X.

**Figure 1** Inductive coupling to nanocrystals linked to DNA and evidence of dehybridization. (a) Sequence of the hairpin molecule, M. The molecule is self-complementary at the ends for 7 bases, with a primary amine in the loop to which a 1.4-nm gold nanocrystal is covalently linked. (b) Absorbance at 260 nm ($A_{260}$) of a solution of M in RFMF (squares). Arrows indicate when the RFMF is on/off. Circles, response of N (no nanocrystals) in RFMF. Gold nanocrystals also show an absorbance increase under RFMF, qualitatively attributed to the change in the optical absorption due to eddy currents. However, the absorbance increases nearly uniformly over 200–300 nm with RFMF and is small relative to the oligo change in absorbance. To obtain the data shown here, the nanocrystal contribution was subtracted (see Supplementary Information for details). M with a fluorophore/quencher pair on the ends shows fluorescence increasing reversibly with RFMF (not shown), confirming dehybridization.
photometer as the RFMF was pulsed at 15-s intervals (Fig. 1b). As the RFMF was switched on, $A_{260}$ increased from 0.22 to 0.25, indicating that $M$ dehybridized with the RFMF (squares in Fig. 1b). When the RFMF was switched off, $A_{260}$ returned to its original value. Cycling through the on and off states was repeatable. The on/off ratio of $A_{260}$ is 1.09, consistent with calculated absorption coefficients for this sequence hybridizing and dehybridizing 7 bases. Control solutions with DNA only (that is, not linked to gold nanocrystals), referred to as $N$, resulted in no change in the absorbance with RFMF (circles in Fig. 1b). This shows that inductive coupling to a covalently bound metal nanocrystal can reversibly dehybridize DNA on a timeline of at most several seconds.

To determine the effective temperature that inductive coupling to the nanocrystal produces locally, we designed a two-phase system in which DNA is dehybridized from a solid support into solution (Fig. 2a). $X$ is a 12-nucleotide DNA molecule with a nanocrystal on the 3’ end and a fluorophore (FAM) on the 5’ end (see Methods). It was hybridized to its complement $Y$, which was immobilized onto 100-μm streptavidin-coated agarose beads comprising the solid phase. As $X$ dehybridizes from $Y$, it diffuses into the supernatant. The amount of dehybridization is proportional to the amount of $X$ in the supernatant, which was measured by fluorescence spectroscopy. One aliquot of the sample was exposed to the RFMF and its supernatant removed. Its fluorescence spectrum was compared to supernatants of aliquots thermally heated at specific temperatures. The intensity of the supernatant fluorescence spectra (Fig. 2b) increases with temperature, indicative of increased amounts of $X$ dehybridized. The sample exposed to RFMF has intensity (solid line) in between the 30 °C and 50 °C samples. The integrated peak intensity of the spectra (Fig. 2c) as a function of temperature (squares) can be fitted to a sigmoidal (solid line), corresponding to a thermal denaturation curve. The intensity of the RFMF sample (dashed line) extrapolates to 35 °C, indicating that the effective temperature increase on $X$ produced by the RFMF is +13 °C. This change in temperature is sufficient for control of many biological processes.

One important property is the ability to address molecules with a nanocrystal while having a lesser effect on molecules not bound to a nanocrystal. If induction heating of the nanocrystal is sufficiently spatially localized, it will afford selectivity. We used a two-phase system in which $X$ was mixed with $Z$, identical in sequence to $X$ but instead has TMR (see Methods) on the 3’ end (Fig. 3a). TMR emits at a wavelength distinct from FAM ($\lambda_{\text{max}} = 563$ nm). The two-phase system comprises tetrameric avidin acrylic beads with both $X$- and $Z$- $Y$ hybrids on the surface in approximately equimolar amounts. One sample was exposed to RFMF, and the supernatant was compared to a thermally heated sample (70 °C) by fluorescence. The difference of the spectrum before and after dehybridization for the 70 °C sample (Fig. 3b, right) has peaks at 515 nm and 563 nm, indicating dehybridization of both $X$ and $Z$. The RFMF sample (Fig. 3b, left) shows intensity at 515 nm due to dehybridized $X$ but negligible intensity at 563 nm, which indicates selectivity: $Z$ is relatively unaffected by induction heating of $X$ in its proximity. Quantification of the spectra yields ~80% $X$ for the RFMF sample and ~55% for the 70 °C sample. The slight amount of $Z$ dehybridized in the RFMF sample is presumably due to the proximity of the molecules associated with tetrameric avidin (intermolecular separation is expected to be ~10 nm).

RFMF experiments on solutions of mixtures of $M$ and $N$ also illustrate selective heating. Figure 3c shows $A_{260}$ with RFMF on and off for a fixed concentration of $M$ as $N$ was added. Values of $A_{260}$ for the on and off states increase linearly, but the difference ($\Delta A_{260}$) remains constant. Samples of increasing concentration of $M$ show $\Delta A_{260}$ increasing with concentration (not shown). These experiments indicate that induction heating of the nanocrystal on $M$ is sufficiently localized that $N$ is unaffected. Another indication of selectivity of RFMF dehybridization is evident in the purification of gold-linked $X$ from unlabelled $X$, where the fraction of gold-linked oligonucleotides increases with successive RFMF dehybridizations. Figure 3d shows the fluorescence spectra of a sample before and after RFMF purification. A sample of partially labelled $X$ is RFMF dehybridized and compared to a 70 °C aliquot, which effectively provides the total number of $X$ in the system (dots). If the RFMF supernatant is rehybridized to $Y$ and RFMF dehybridized a second time, its intensity is higher relative to the 70 °C aliquot (lines). The percentage of molecules dehybridized by RFMF increases after purification, suggesting the RFMF preferentially dehybridizes $X$ linked with nanocrystals. If RFMF dehybridization were non-selective, this percentage would be the same for both samples. These experiments show that induction heating of the nanocrystal is localized such that surrounding molecules are not substantially affected.

Manipulation of DNA by itself is useful, as it has potential as an...
actuator and performing computations. Because nanocrystals can be readily attached to proteins, switching more complex processes—such as enzymatic activity (S.Z., J. Shi, M. Jura, K.H.-S., J.J.S. and J.M.J. manuscript in preparation), biomolecular assembly, gene regulation, and protein function—might be possible. Due to the spatial localization of denaturation, it might be possible to control portions of proteins or nucleic acids while the rest of the molecule and neighbouring species would remain relatively unaffected. Control of dehybridization of antisense oligos in transcription would permit reversible control of the production of specific messenger RNA. Because the addressing is not optical, this technology could be useful in highly scattering media.

Methods

Linking gold nanocrystals to DNA hairpin-loop molecules

DNA oligonucleotides were synthesized via solid-phase synthesis (Research Genetics) with addition of internal amine groups, terminal biotin, or fluorophores. DNA hairpin-loop oligonucleotides of the type used to make M (‘molecular beacons’15) have a thymine in the loop region which was modified on the 5′ position of the base with a C6 primary amine, to which a nanocrystal was covalently linked. 1.4-nm uncharged gold nanocrystals functionalized with a single sulfo-N-hydroxy-succinimide (NHS) ester (Nanoprobe) were incubated with the oligonucleotide in the manufacturer’s buffer at room temperature for 1 h, resulting in a covalent bond between the nanocrystal and oligonucleotide. Nanocrystals were in molar excess (>10:1). Excess nanocrystals were removed by 70% ethanol precipitation on ice and repeated washes. The nanocrystals are soluble in ethanol, but the DNA or species attached to it precipitates. Gel electrophoresis is effective in separating nanocrystals from DNA-labelled gold nanocrystals, but for short DNA the mobility shift is expected to be too small to resolve the two species. The concentration of M and N is 1 × 10−6 M in 1X PBS (1X PBS = 1 mM KH2PO4, 10 mM NaH2PO4, 137 mM NaCl, 2.7 mM KCl).

RFFM

Alternating magnetic fields were generated by applying an alternating current to a coil with 35 turns and a cross section of ~1 cm2. Coils were wrapped around a plastic cuvette/tube holder with open structures to maximize light passage. Currents with f = 1 GHz were obtained using an RF signal generator (Hewlett Packard 8648C) with an output of 1 mW in conjunction with a linear amplifier. The ultimate output power range used was 0.4–4 W, though these are upper limits to the exact power inside the coil due to losses from set-up architecture (10% input power estimated to be transferred to coil). DNA hairpin samples (volume 200 μl) were in a 3 mm × 3 mm quartz cuvette inside the coil. The two-phase samples were in 250 μl plastic tubes (supernatant volume 165 μl, solid phase volume 60 μl) put into the coil in a water bath. The power used for the experiments shown in Figs 1 and 2 was 4 W, and 1 W for those in Fig. 3.

Spectroscopy

UV–visible spectra were taken on a DU530 Beckman spectrophotometer with time resolution limited to 5 ± s steps. Fluorescence spectra was taken on a Spex Fluoromax fluorometer with the RFFM, the supernatants were separated from the cuvette (entrance/exit slits 1–5 mm, λex/emax 450 nm). Signal was averaged multiple times (≤10). Fluorescence peak areas were quantified by Yoigt lineshape deconvolution using a commercial program (Peakfit, SPSS). Figure 3b fits had the same parameters for peak positions and widths for both spectra.

Two-phase system

Molecule X is a 12-nucleotide DNA that has a 6-carboxyfluorescein (FAM, Amax = 516 nm) on the 5′ end and a C6 primary amine on the 3′ end (Tmax = 40 °C). It was labelled with the nanocrystal as described above. Molecule Y is a 68-nucleotide DNA complementary to X, and has a biotin on the 5′ end. X and Y were hybridized, and the X-Y hybrid was incubated with streptavidin–agarose beads (Sigma-Aldrich) with <d> = 100 μm to allow the biotin on Y to bind to streptavidin. Samples were washed several times with 1X PBS to remove free gold. This biotin–streptavidin interaction (Kd = 10−11 M) is robust for the temperatures used here.16 Tests in which bead-bound Y was heated to 85 °C showed no presence of oligonucleotide in the supernatant, confirming the biotin–streptavidin interaction intact.

In the construction of the two-phase system (Fig. 2), X was purified from unlabelled oligonucleotides by exposing bead-immobilized X-Y to RFFM. Oligonucleotides not linked to a nanocrystal had no response to RFFM and thus remain hybridized to Y. This is another alternative to purification by gel electrophoresis. Purified X was hybridized to Y, then exposed to streptavidin–agarose beads to immobilize X-Y. After multiple washes with 1X PBS, buffer was added to the beads to serve as the liquid phase. Samples were exposed to RFFM or heated conventionally at a specific temperature for 5 min. Following heating or exposure to the RFFM, the supernatants were separated from the beads and then measured by fluorescence spectroscopy. All fluorescence spectra were taken under identical conditions (temperature, spectrometer parameters).

Mixed two-phase system

Oligonucleotide Z was identical in sequence to X but was labelled on the 3′ end with NHS-tetramethylrhodamine (TMR, Molecular Probes). Attachment of the TMR to the primary amine on Z was achieved by incubating the molecules together in sodium bicarbonate solutions (1 h, room temperature) in the dark to prevent photobleaching (manufacturer directions). Oligonucleotide Z was hybridized to Y, mixed in solution with X-Y, then incubated with tetrameric avidin acryl beads (Sigma-Aldrich) as described above. Free molecules were removed with multiple washes of 1X PBS. The sample is equimolar in X to Z on the beads. The surface density of the monomeric avidin on the beads is ~1 molecule per nm2.

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Supplementary Information

The paper appears on the Nature website (http://www.nature.com).

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